

1 **Photocatalytic activity of CuO and Cu₂O nanowires**

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10 **Abstract**

11 Here we report the photocatalytic efficiency of CuO and Cu₂O nanowires studied by dye
12 degradation in water. The CuO nanowires were synthesized on Cu foils by thermal oxidation. A
13 subsequent thermal annealing induced a controllable phase transformation of the nanowires and the
14 underlying film from CuO to Cu₂O phase. A structural characterization was made by scanning and
15 transmission electron microscopies, energy dispersive x-ray analysis and x-ray diffraction. The
16 photocatalytic performances were evaluated by the degradation of two dyes in water: methylene
17 blue and methyl orange. In particular, CuO nanowires showed a good photocatalytic activity in
18 degrading methyl orange, compared with Cu₂O nanowires. The photocatalytic activity was
19 correlated to the physical-chemical properties of the samples, revealing the detrimental effect of the
20 polycrystalline structure for photocatalytic process.

21

22 **1. Introduction**

23 Cupric oxide (CuO) and cuprous oxide (Cu₂O) are P-type semiconductor with a narrow band
24 gap of 1.2 eV and 2 eV, respectively. They received much attention for their numerous applications
25 in several fields [1-4]. In particular, CuO has been extensively studied because of its close
26 connection to high-*T_c* superconductors [5], but it has also been widely exploited as a powerful

27 heterogeneous catalyst to completely convert hydrocarbons into carbon dioxide and water [6]. In
28 addition, CuO and Cu₂O can be potentially used in gas sensors, solar cells, field emitters, electronic
29 cathode materials and catalysts [2,3, 7-11].

30 Several methods were developed to synthesize CuO and Cu₂O with various morphologies,
31 such as nanoparticles by alcoholthermal decomposition of copper acetate [12], nanodendrite and
32 microcrystals via the hydrothermal route [13,14], nanorods and nanoribbons by wet chemical
33 methods [15]. The direct and simple thermal oxidation method was employed to synthesize CuO
34 nanowires and nanorods. Many research teams prepared CuO nanowires successfully by oxidizing
35 copper foils under different conditions of annealing temperatures, times and atmospheres [16,17].

36 Environmental safety has recently become one of the most serious concerns for the scientific
37 community because of rapid increase of organic pollutants used in agriculture and various industries
38 which cause severe adverse effects on the environment. Since photo-induced decomposition of
39 water on TiO₂ electrodes was discovered [18], semiconductor-based photocatalysis has attracted
40 extensive interest, in particular for water purification [19,20]. In this contest, CuO and Cu₂O are
41 good promising candidates for wastewater treatment, because of their photocatalytic properties.
42 However, due to the anisotropy of crystals, Cu₂O exhibits different stabilities, activities, etc. For
43 example, octahedral Cu₂O with exposed {111} facets exhibited much higher photocatalytic activity
44 than cubic Cu₂O [14].

45 In this work, we synthesized CuO nanowires by a thermal process in oxygen ambient of a
46 Cu foil. Nanowires were morphologically and structurally characterized by scanning electron
47 microscopy (SEM), transmission electron microscopy (TEM) and energy dispersive x-ray analysis
48 (EDX). The material shows a high density of CuO nanowires, more than 10 μm in length and ~ 80
49 nm in mean diameter. A subsequent thermal annealing induced a controllable phase transformation
50 of the nanowires and the underlying film from the CuO to Cu₂O. The photocatalytic activity of the
51 synthesized nanowires were examined through the degradation of methyl blue and methyl orange,

52 in agreement with industry accepted international standards (ISO 10678:2010), and it was
53 correlated to the physical-chemical properties of the materials.

54

55 **2. Experimental methods**

56 CuO nanowires were synthesized on Cu foils (purity of 99.9%, 0.4 mm thick) by thermal
57 process in oxygen at the temperature of 550 °C for 3 h. Before of the thermal process, the copper
58 foils were cleaned in acetone and isopropanol alcohol, under ultrasonic bath for 10 min, so to
59 remove possible contaminations. After a drying process in air, the foils were annealed in a
60 conventional furnace under a controlled O₂ flux.

61 The morphology and the chemical composition of the samples were investigated by field
62 emission scanning electron microscopy (FE-SEM Zeiss Supra 35) operated at an acceleration
63 voltage of 5 kV and equipped with energy dispersive X-ray spectroscopy (Aztec-EDS system by
64 Oxford Instruments with 80mm² X-Max SDD detector).

65 The TEM characterization was made using a Jeol 2010F operated at 200 kV, in bright field and
66 diffraction condition.

67 XRD measurements were acquired by Bruker D-500 diffractometer at an angle of incidence
68 of 0.8°, and Θ -2 Θ from 20 to 60°. Acquired spectrum were analyzed by Bruker software suite,
69 including ICSD structure database.

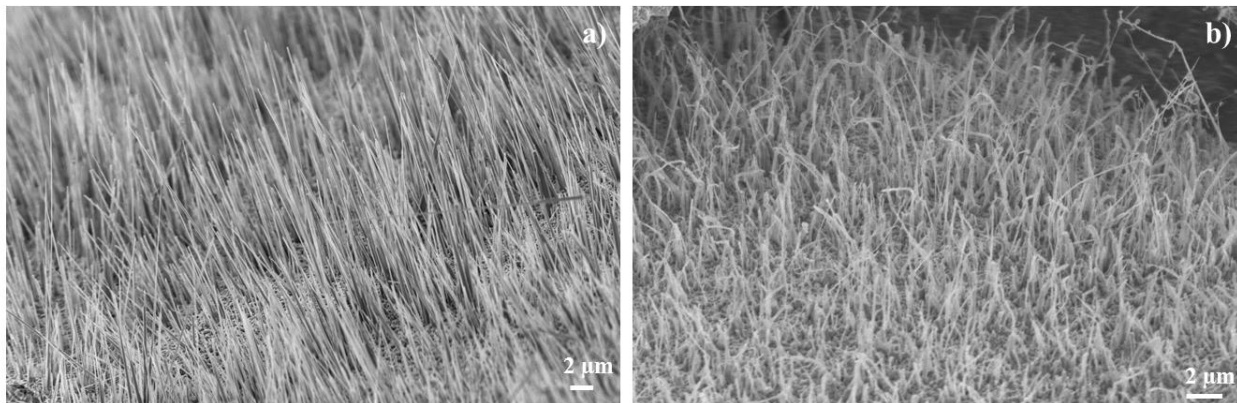
70 The photocatalytic activity of the investigated materials with respect to the degradation of
71 organic compounds in water was evaluated by two colorants Methylene Blue (MB) and Methyl
72 Orange (MO). The MB is a chemical compound commonly used to evaluate the photocatalytic
73 efficiency of the materials, while MO is a toxic dye commonly used in the textile industry. A
74 serious problem for the comparison of experimental results from different laboratories is the fact
75 that every research group is still using its “own” chemical test system for the evaluation of the
76 photocatalytic activities. In this work we use an experimental setup in agreement with the

77 international standards ISO 10678:2010. This protocol requires, like light source, a lamp in the
78 range of 320-400 nm. The samples, 1 cm² in size, were immersed in a solution (2 ml) containing
79 MB or MO and de-ionized water (starting concentration: 1.5×10⁻⁵ M and 1×10⁻⁵ M, respectively).
80 The mixture was irradiated by an UV lamp (350-400 nm wavelength range) with a power of 8 W
81 for 4 hours. Every 30 minutes, for the first two hours (and then every hour), of irradiation the
82 solution was measured with a UV-VIS spectrophotometer (Perkin-Elmer Lambda 45) in a
83 wavelength range between 500-800 nm for MB and 350-650 nm for MO. The degradation of MB
84 and MO was evaluated by the absorbance of the peak at 664 nm and 464 nm respectively, thanks to
85 the Lambert-Beer law ($A = \epsilon \times l \times C$, where A is the absorbance of the solution at 664 or 464 nm,
86 ϵ is the extinction molar coefficient, l is the width of the cuvette, C is the concentration of the MB
87 or the MO) [21]. The decomposition of the colorants in absence of catalyst materials was also
88 checked as a reference. Before the measurements, the samples were irradiated by the UV lamp for
89 50 min in order to remove the possible presence of hydrocarbons on the sample surface [22].

90

91 **3. Results and discussion**

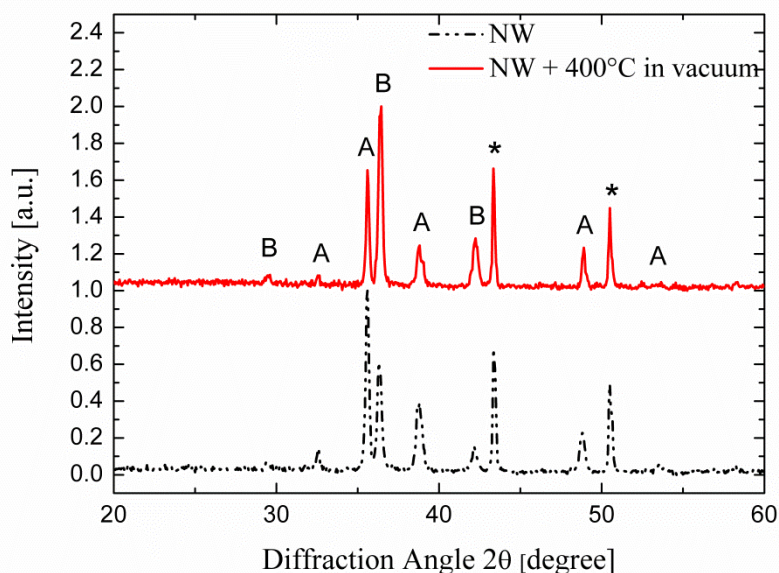
92 Figure 1 (a) reports SEM images of nanowires (NWs) grown with high density by oxidation
93 of a Cu sheet for 3 h at 550°C. The growth of the nanowires was preceded by a deformation of the
94 substrate due to a straining force, with the formation of a hill and valley structure, in accordance
95 with the literature [23]. SEM analyses in cross-view (not reported here) allowed to determine a
96 mean length of the NWs of 10 μm and a mean diameter of ~ 80 nm. Figure 1(b) reports the SEM
97 image of the NWs after a subsequent thermal annealing in vacuum for 3 h at 400 °C. They appear
98 bent compared to the as-grown NWs.



99

100 **Fig. 1:** Plan view SEM images of nanowires growth on copper foil: (a) by thermal process in oxygen at 550°C for 3h,
 101 (b) after a thermal annealing in vacuum at 400°C for 3h.
 102

103 The chemical composition was investigated by XRD and EDX analysis. In particular, Figure
 104 2 shows the XRD spectra of the NW before and after a subsequent thermal treatment in vacuum at
 105 400 °C for 3 h. The mixture phase of CuO and Cu₂O can be detected in both samples. In more
 106 detail, before of the annealing (dotted line in Fig. 2) CuO is the prevailing phase (indicated with
 107 “A” in the figure), after the annealing (continuous line) the prevailing phase is Cu₂O (indicated with
 108 “B” in the figure). The signals indicated with the asterisks came from the substrate (Cu foil).



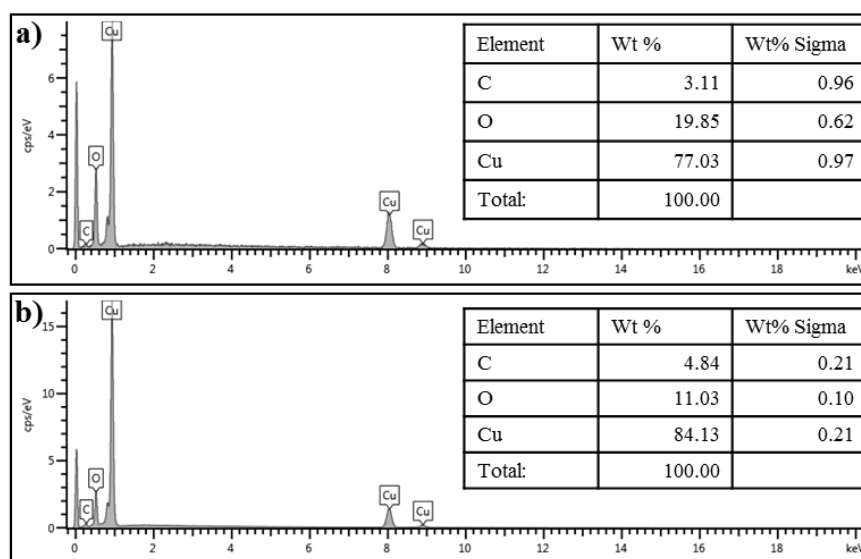
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110 **Fig. 2:** XRD patterns of the NWs before (dotted line) and after the annealing (continuous line). “A” refers to CuO
 111 phase, “B” refers to Cu₂O phase, “*” refers to Cu.
 112
 113

114 Figure 3 reports the EDX spectra collected by SEM in cross-section, on the surface of the
 115 nanowires before (Fig. 3 (a)) and after (Fig. 3 (b)) the thermal treatment in vacuum. Only the
 116 presence of Cu and O is detected, in addition to adventitious carbon. In the inset of the EDX
 117 spectra, we report the tables of the detected elements with the relative percentage in weight (Wt%).
 118 From these percentages was possible to determine the stoichiometry of NWs by the equation:

$$119 \quad \%_x = \frac{PM_x}{PM_{\text{compound}}} \times 100 \quad (1)$$

120 where $\%_x$ is the relative percentage in weight of the x element; PM_x is the molecular weight of the x
 121 element; PM_{compound} is the molecular weight of the compound. Adventitious carbon is not
 122 considered in the calculation. In particular, the as-grown NWs are composed of CuO. After the
 123 thermal treatment in vacuum, the chemical composition changes and it moves to Cu₂O phase.

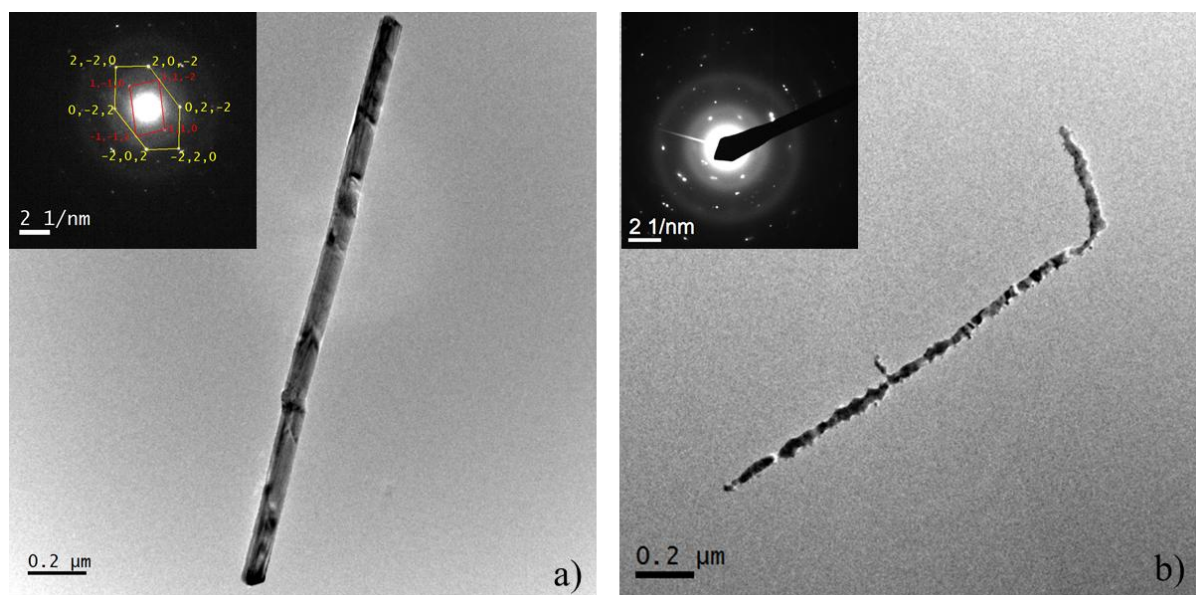


124

125 **Fig. 3:** EDX spectra with the tables of the detected elements, for: (a) nanowires growth by thermal process in oxygen at
 126 550°C for 3h, and (b) after a thermal annealing in vacuum at 400°C for 3h.
 127

128 Figure 4 reports TEM images of CuO (Fig. 4 (a)) and Cu₂O nanowires (Fig. 4 (b)). The
 129 TEM characterization shows the mono-crystalline structure of CuO-NWs, detecting few defects. It
 130 was confirmed by a diffraction analysis on a selected area, as reported in the inset of Fig. 4 (a). We
 131 can observe a 111 orientation of the CuO mono-crystal which was confirm by Carine simulation.

132 Instead, Cu_2O -NWs appeared poly-crystalline, as confirmed by the diffraction (inset Fig. 4(b)) and
133 dark-field characterizations (not reported here).



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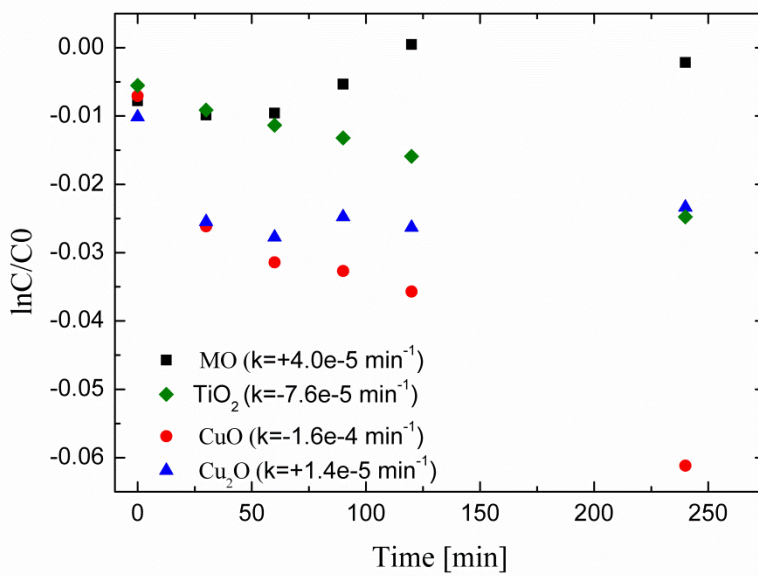
135 **Fig. 4:** TEM analysis of: (a) mono-crystalline CuO nanowires, (b) poly-crystalline Cu_2O nanowires. The diffraction
136 patterns are reported in the insets.

137

138 The ability of the synthesized material in degrading organic pollutants was tested with MB
139 and with MO. Methyl orange, is a toxic dye compound commonly used in textile, leather tanning
140 and paper production industries. In detail, azo dyes represent about 50-70% of the waste dyes
141 produced by the industries [24], and about 10-15% of them comes from the textile industry [25].
142 CuO and Cu_2O nanowires did not show a significant efficiency in degrading the MB (the results are
143 not reported here). This depends on the nature of the MB and NWs: MB is a cationic dye, and CuO
144 and Cu_2O are a P-type semiconductor [26,2]. Therefore, the absorption of the compound on the
145 surface of the nanostructures is hindered because of the electrostatic repulsions between the positive
146 molecular charge of the MB and the P-type semiconductor. This confirms that the nature of the
147 organic compound and the nanostructures is fundamental in the first step of a photocatalytic
148 process, that is the adsorption. This is not the case of MO, since it is an anionic dye.

149 Figure 5 reports the logarithmic residual concentration of MO under UV light, versus the
150 irradiation time. In particular, C is the concentration of the MO after the irradiation, C_0 is the
151 starting concentration of the MO.

152



153

154 **Fig. 5:** MO degradation under UV light for three samples: MO (squares), MO with TiO₂ thin film (diamond), MO with
155 CuO-NW (circles), MO with Cu₂O-NW (triangles).

156

157 We tested four samples: MO in absence of any catalyst materials (squares), MO with TiO₂ thin film
158 (diamond), MO with CuO-NWs (circles), MO with Cu₂O-NWs (triangles). TiO₂ film was deposited
159 on Si substrate, with a Picosun R-200 advanced system. The film thickness (~10 nm) was evaluated
160 by the M-2000 spectroscopic ellipsometer by Woollam. This film was used as reference sample.

161 Preconditioning process was applied to remove possible adsorbed organic pollutants on the
162 surfaces [see the Experimental Method section]. The samples were immersed in the MO solutions
163 and kept in the dark more than one hour. This process allows evaluating the absorption of the MO
164 on the different sample surfaces without involving any photocatalytic reaction. However nanowires
165 do not show absorption. After this step, the samples were exposed to the UV light, so to induce the
166 photocatalysis. The observed reaction rate, k , follows a Langmuir-Hinshelwood model kinetics,
167 which can be expressed by a first-order reaction kinetic:

$$\ln\left(\frac{C}{C_0}\right) = -kt$$

168

169 where C is the concentration of organic species, C_0 is the initial concentration of organic species, t
170 is the irradiation time [27]. The values are reported in Figure 5 for all the analyzed samples. For
171 MO and Cu₂O-NWs, k has a positive value, due to the evaporation of the solvent content in MO
172 solution and due to the incapability of the Cu₂O-NWs in degrading the MO. Consequently, there is
173 an increase of the MO concentration in the solutions increasing the exposition time. While for TiO₂
174 thin film and CuO-NWs, the k value is negative, due to the ability of the samples in degrading MO.
175 In particular, CuO-NWs shows the best photocatalytic activity.

176 It is reported in literature that the reaction rate, k , depends linearly on the amount of the
177 photocatalytic material present in the solution [28]. However, it is difficult to make a comparison
178 with the reaction rate values reported in the literature, because of the wide range of experimental
179 conditions used, such as: power lamp, spectral range, dye concentration, presence of hydrogen
180 peroxide in solution to increase the photocatalytic ability of the material [29,30].

181 In comparison with TiO₂ thin film, Cu⁺ promoted electron-hole generation due to the lower
182 band gap, and the existence of copper is beneficial in transferring photo induced electrons and
183 enhancing the separation efficiency of photo-induced electron hole pairs, which can reduce the rate
184 of charge-carrier recombination. Consequently, the rate of the primary interfacial charge transfer
185 increases [31]. The different response of CuO and Cu₂O nanowires in the degradation of MO
186 probably depends on the different structure of the wires: mono-crystalline for CuO and poly-
187 crystalline for Cu₂O-NWs (as revealed by diffraction analyses). In addition, the transformation of
188 the nanowires and the underlying film from the CuO to Cu₂O phase introduces many defects in the
189 structure, as revealed by the TEM analyses (Fig. 3(b)). It is well known that defects in TiO₂ play a
190 crucial role in the photocatalytic properties of the titanium dioxide. In particular, the nature of
191 defects is fundamental because they can on one hand increase the photocatalytic efficiency (i.e.
192 oxygen vacancies) [32] or, on the other hand, they can act as recombination centers for electrons

193 and holes [33]. Moreover, poly-crystalline Cu₂O-NWs expose different facets, that have different
194 photocatalytic activity [34].

195

196 **4. Conclusion**

197 In this work, we synthesized CuO nanowires by a simple thermal process of Cu foils in a
198 controlled oxygen ambient. XRD, EDX and TEM analysis confirmed that a subsequent thermal
199 treatment in vacuum is able to modify the chemical structure of the nanowires and of the underlying
200 film from CuO to Cu₂O. The synthesized CuO-NWs revealed a photocatalytic activity in the
201 degradation of methyl orange higher than the one of Cu₂O-NWs. The different response probably
202 depends on the large amount of defects induced by the thermal treatment in vacuum due to the
203 formation of a polycrystalline structures. The importance to study and obtain high aspect-ratio
204 nanostructures and, in future, to join the investigated material with different semiconductors should
205 be evaluated to try to increase the photocatalytic efficiency in degrading organic pollutants.

206

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210

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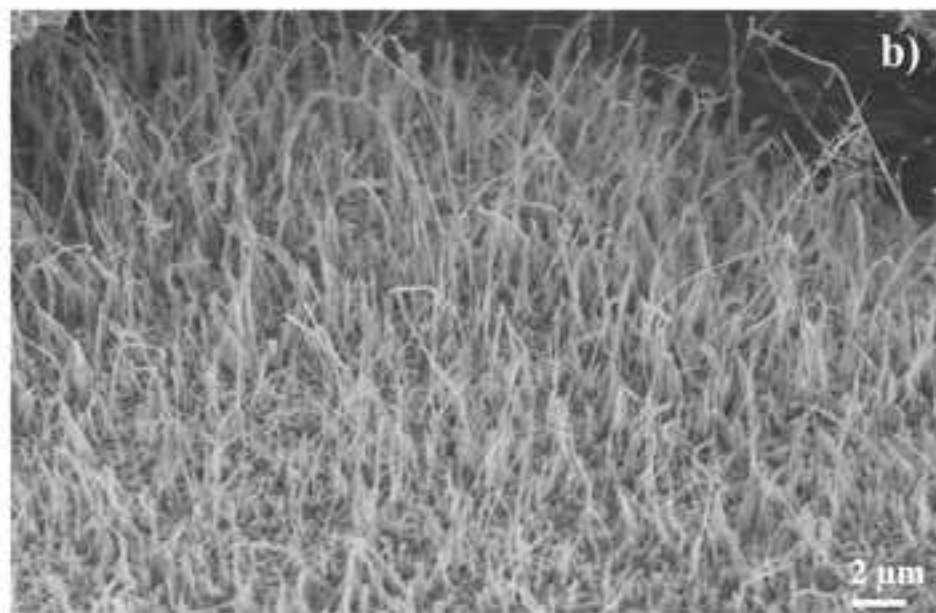
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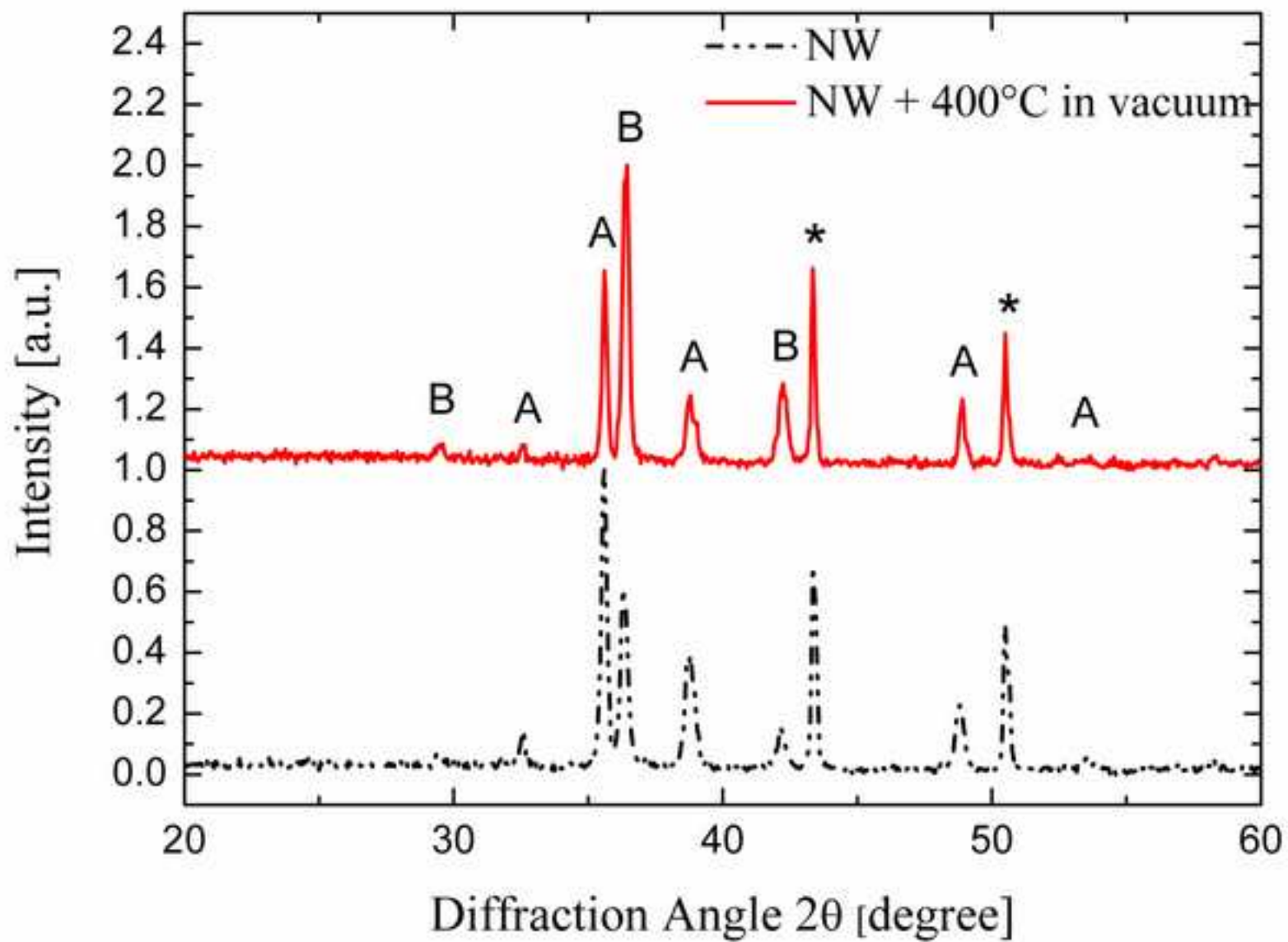
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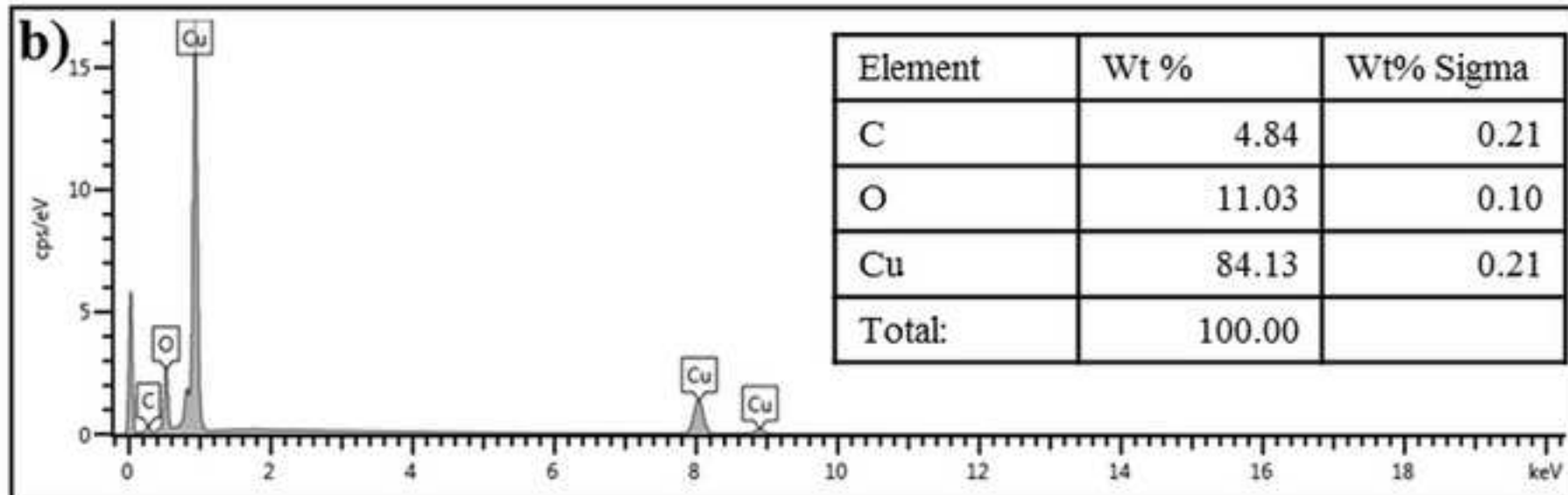
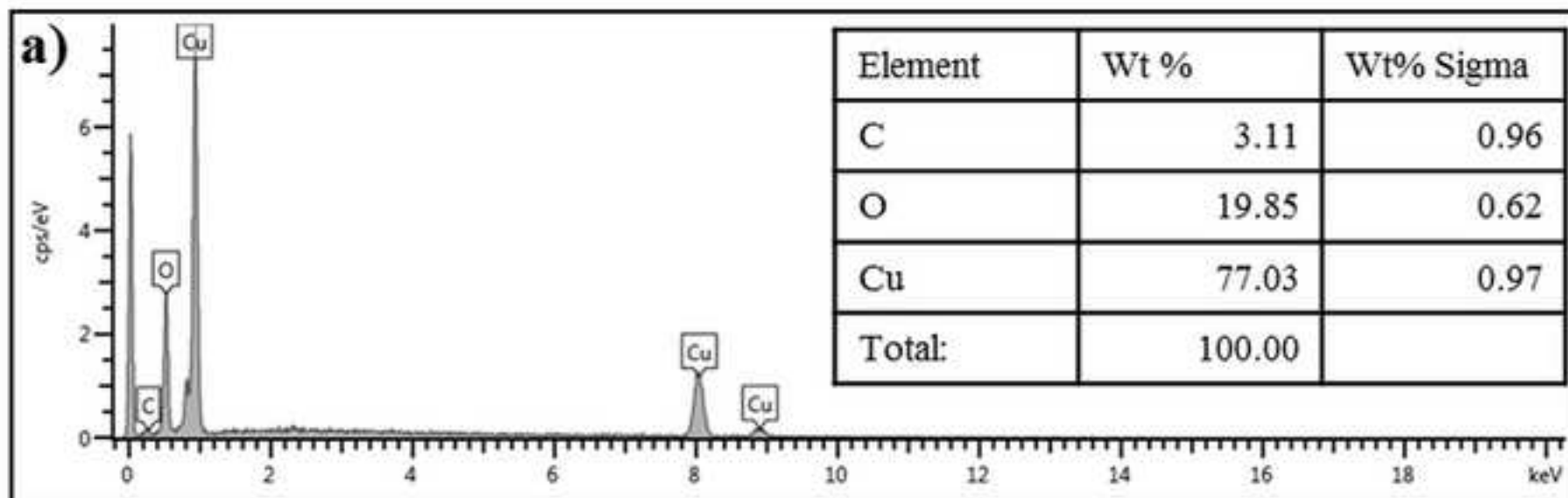
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